



## ORIGINAL ARTICLE

# Surface-reconstructed Cu electrode *via* a facile electrochemical anodization-reduction process for low overpotential CO<sub>2</sub> reduction



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## KEYWORDS

CO<sub>2</sub> reduction;  
Cu electrode;  
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**Abstract** A high-surface-area Cu electrode, fabricated by a simple electrochemical anodization-reduction method, exhibits high activity and selectivity for CO<sub>2</sub> reduction at low overpotential in 0.1 M KHCO<sub>3</sub> solution. A faradaic efficiency of 37% for HCOOH and 27% for CO production was achieved with the current density of 1.5 mA cm<sup>-2</sup> at -0.64 V vs. RHE, much higher than that of polycrystalline Cu. The enhanced catalytic performance is a result of the formation of the high electrochemical active surface area and high density of preferred low-index facets.

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## 1. Introduction

Several metal electrocatalysts have demonstrated sufficient capability for CO<sub>2</sub> reduction in aqueous solutions [1]. Among them, Cu has been identified to catalyze the electroreduction of

CO<sub>2</sub> to hydrocarbons [2–4], although high overpotentials are in general required with poor selectivity. Various strategies have been attempted to improve the catalytic performance of Cu including the control of macrostructure [5], particle size [6], exposed surface facets [7], and the formation of alloys with secondary metals [8], but the selective CO<sub>2</sub> reduction at low overpotential remains a huge challenge. A novel strategy was recently developed by Kanan et al. to produce a nanostructured Cu electrode from its oxides [9,10], which shows a Faradic efficiency of 40% toward CO at -0.5 V vs. RHE, with a total current density of 2.7 mA/cm<sup>2</sup> [9]. However, a high temperature and a long reaction time (500 °C for 8–12 h) are typically required for the preparation of above oxide-derived Cu (OD-Cu) catalysts. There is still a continuing need for easily accessible Cu catalysts for CO<sub>2</sub> reduction in an energy- and

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time-efficient manner. It has been reported that the electrochemical oxidation and reduction is an effective method to prepare high active Ag electrocatalysts for CO<sub>2</sub> reduction [11,12]. Since Cu has an extensive redox electrochemistry based on oxidation to Cu<sup>2+</sup> and reduction to Cu<sup>1+</sup> and Cu<sup>0</sup> [13,14], it is highly desirable to explore the Cu electrochemistry to generate efficient catalysts for the CO<sub>2</sub> reduction reaction by a solution-based method. Herein, we report the preparation of a Cu electrocatalyst with a high-surface area for efficient CO<sub>2</sub> reduction *via* a facile anodization-reduction process.

## 2. Experimental

For the preparation of highly dense Cu(OH)<sub>2</sub> nanowires (NWs) on Cu foil substrate, pieces of Cu foil (1.5 × 1.0 cm<sup>2</sup>) were first washed with 1 M HCl solution for several seconds to remove the native oxide layer. After that, Cu foil was electrochemically anodized in 1 M NaOH solution at a current density of 2 mA cm<sup>-2</sup> in a three-electrode system with a Pt foil (2.0 × 1.0 cm<sup>2</sup>) and a saturated Ag/AgCl used as counter and reference electrode, respectively. The morphologies of samples were observed by an FEI Quanta 600 Scanning Electron Microscope (SEM) and an FEI Titan ST Transmission electron microscopy (TEM). The X-ray diffraction (XRD) patterns were recorded on a Bruker model D8 Advance. X-ray photoelectron spectroscopy (XPS) was conducted using a Kratos Axis Ultra DLD spectrometer. All the electrochemical experiments were carried out with a Biologic VMP-300 potentiostat in an air-tight and glass frit-separated two compartment three-electrode electrochemical cell. For the bulk electrolysis of CO<sub>2</sub>, the cathodic compartment of the cell was degassed and saturated with CO<sub>2</sub> at a rate of 10 mL min<sup>-1</sup>. The eluent was delivered directly to the sampling loop of an on-line gas chromatograph (Agilent 7890B). The liquid product HCOOH was analyzed using a high-pressure liquid chromatography (HPLC, Agilent technologies) system.

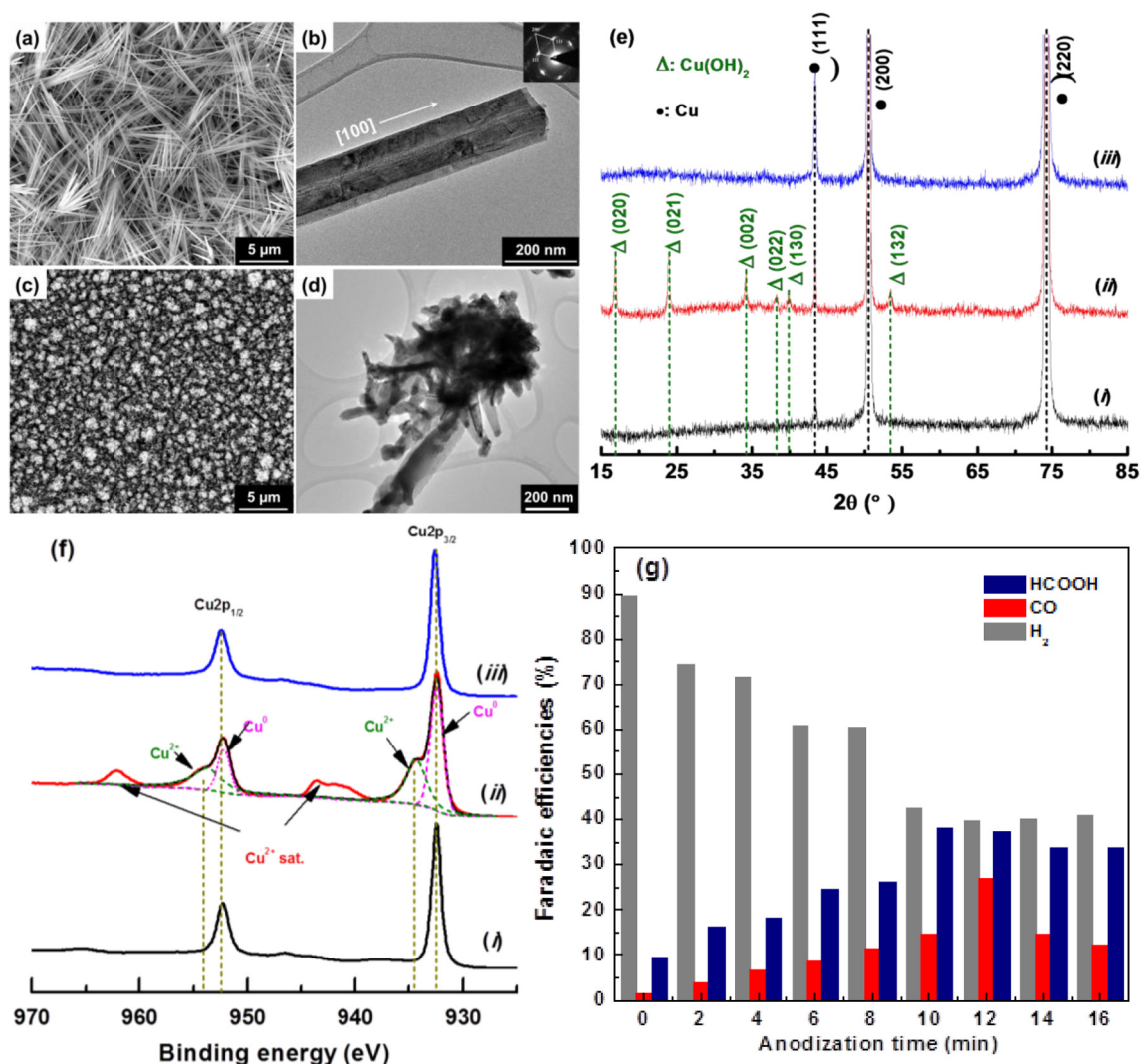
## 3. Results and discussion

SEM image (Fig. 1(a)) of a Cu electrode anodized at 12 min showed that the anodized Cu foil was entirely covered by nanowires with diameters in the range of 100–250 nm and an average length of 4 μm. XRD and XPS analysis revealed that the product was dominated by Cu(OH)<sub>2</sub> and Cu phases (Fig. 1(e) and (f)) [15], indicating that the growth of Cu(OH)<sub>2</sub> NWs on the surface of Cu foil. During the course of our study, a two-step synthesis of Cu(OH)<sub>2</sub> and CuO nanowire arrays on Cu foil and the electrochemical CO<sub>2</sub> reduction were also reported with the focus on hydrocarbon formation. [16] TEM image of the products (Fig. 1(b)) clearly indicated their nanowire structure with a thickness smaller than 10 nm, while the SAED pattern gave their [100] orientation. High-resolution TEM (HRTEM) image (Fig. S1(a)) showed the formation of plentiful cavities on Cu(OH)<sub>2</sub> NWs. The density and average length of Cu(OH)<sub>2</sub> NWs can be easily controlled by adjusting the anodization time. The XRD patterns in Fig. S2 showed that the intensity for the characteristic peaks of Cu(OH)<sub>2</sub> increased sustainably with the anodization time while the SEM images (Fig. S3) indicated that the length of NWs increased linearly from ~1.75 to 7.0 μm as the anodization time increased.

High-surface-area Cu electrodes were then obtained by subsequently reducing the as-prepared Cu(OH)<sub>2</sub> NWs under the CO<sub>2</sub> reduction conditions (denoted as “anodized Cu” hereafter). Anodized Cu electrode exhibited a flower-like structure with aggregated nanoparticles formed on the tips of the wires (Fig. 1(c) and (d)), a direct result of the surface reconstruction. The HRTEM image in Fig. S1(b) also clearly indicated the formation of grain-boundaries between the nanocrystals, similar to the Cu electrode derived from Cu oxides [9]. XRD patterns and XPS spectrum collected after the electrochemical reduction only provided peaks of metallic Cu, suggesting that the Cu<sup>0</sup> surface was responsible for the following observed activity for CO<sub>2</sub> reduction.

The polycrystalline Cu (poly-Cu) electrode exhibited poor CO (1.5%) and HCOOH (9%) selectivity with almost 90% FE for H<sub>2</sub> evolution (Fig. 1(g)) at applied potential of -0.64 V *vs.* RHE. In contrast, CO and HCOOH selectivities on the anodized Cu electrodes were enhanced, while the H<sub>2</sub> FE was suppressed. Specifically, the CO selectivity continued to increase with the anodization time from 2 min up to 12 min, while the HCOOH selectivity initially increased and reached a plateau of 34–38% FE beyond 8 min. The total FE for CO (27%) + HCOOH (37%) reached a maximum value of 64% at 12 min, where the H<sub>2</sub> FE was below 40%. Meanwhile, the anodized Cu electrodes showed a higher catalytic geometric current density ( $j_{\text{gcd}}$ ) for the CO<sub>2</sub> reduction as compared to poly-Cu and the  $j_{\text{gcd}}$  increased rapidly with the anodization time and reached a maximum at 8 min (2.5 mA cm<sup>-2</sup>), and then became constant in the range of 10–16 min (1.4–1.8 mA cm<sup>-2</sup>) (Fig. S4). Since the anodized Cu electrode possesses nanostructured surfaces, the clearly enhanced  $j_{\text{gcd}}$  and FE could be ascribed to the increased number of catalytically active sites. The electrochemically active surface (ECSA) of anodized Cu electrodes determined by measuring the double-layer capacitance (Fig. S5 and Table S1) is one order of magnitude higher than that of poly-Cu and increased linearly with the anodization time, supporting the positive correlation between  $j_{\text{gcd}}$  and FE and active sites (Fig. S6).

The anodized Cu electrode was also identified to selectively reduce CO<sub>2</sub> at lower overpotentials. Notably, the anodized Cu exhibited appreciable CO (~9%) and HCOOH (~5%) production started at ~-0.3 V *vs.* RHE (Fig. 2(a)), while poly-Cu was totally inert at this potential. When anodized Cu electrode was polarized to more negative potentials, the FEs for CO and HCOOH could be enhanced obviously and a peak FE for HCOOH and CO is obtained at -0.64 V *vs.* RHE. It has been noted that the formation of higher hydrocarbons can be detected on both poly-Cu and anodized Cu electrodes at much negative applied potentials. More specifically, total FEs of 11.3 and 33.1% for CH<sub>4</sub> and C<sub>2</sub>H<sub>4</sub> production were obtained on poly-Cu electrode at -0.941 and -0.996 V *vs.* RHE, respectively, whereas the anodized Cu electrode produces trace C<sub>2</sub>H<sub>4</sub> at all the applied potentials. These results are consistent with the previous observations on the Cu<sub>2</sub>O-derived Cu catalyst [9], indicating that the surface structures of Cu(OH)<sub>2</sub>-derived Cu catalyst though an electro anodization-reduction route are distinct from those of the polycrystalline Cu. On the other hand, the anodized Cu electrode has higher  $j_{\text{gcd}}$  than the poly-Cu electrode in the potential range of -0.3 to -1.0 V *vs.* RHE (Fig. 2(b)), which is associated to its high surface area.

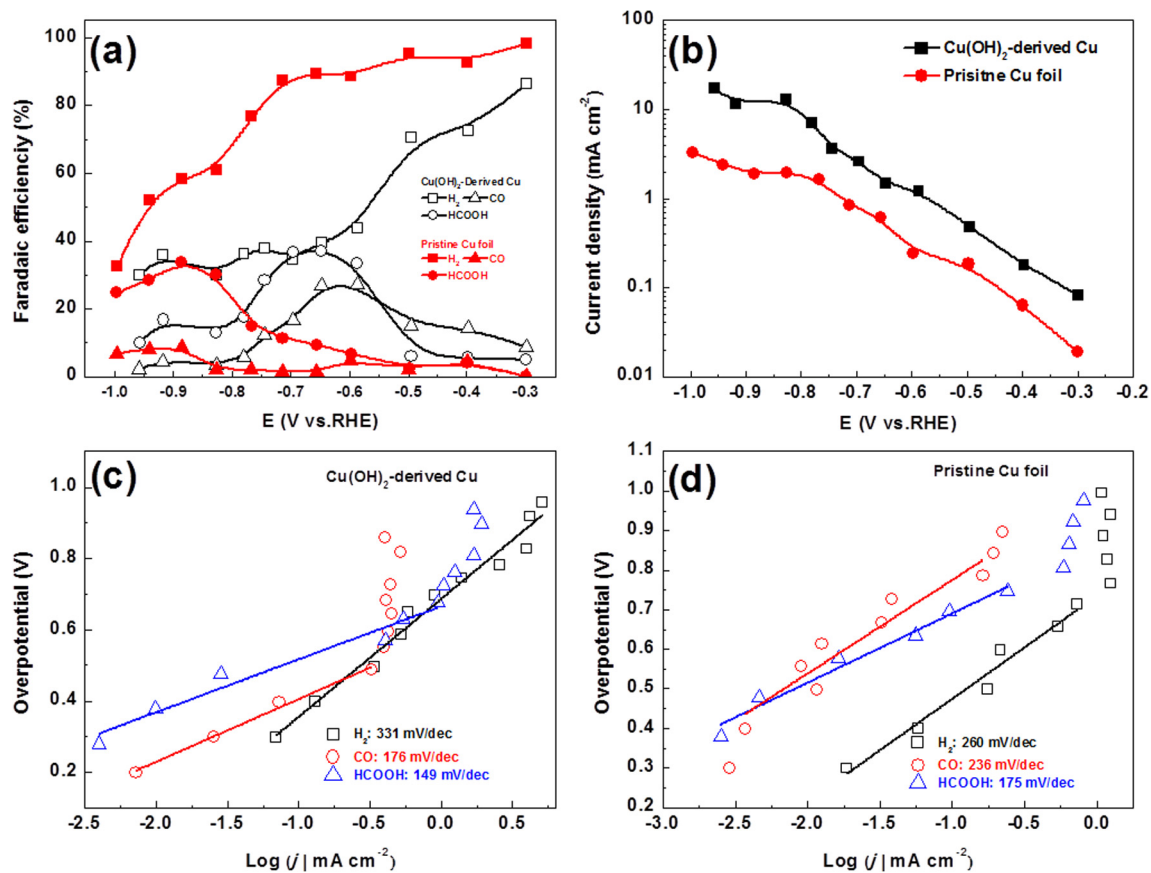


**Fig. 1** SEM, TEM images of anodized Cu electrode (12 min) (a,b) and the same electrode after CO<sub>2</sub> reduction electrolysis at  $-0.64$  V vs. RHE (c,d). (e) XRD patterns and (f) XPS spectra of the poly-Cu (i), the anodized Cu electrode (12 min) (ii), and the anodized Cu electrode after electrochemical reduction in CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> solution at  $-0.64$  V vs. RHE (iii). (g) FEs of H<sub>2</sub>, CO, and HCOOH for poly-Cu and those anodized at different times in CO<sub>2</sub>-saturated 0.1 M KHCO<sub>3</sub> at  $-0.64$  V vs. RHE (with experimental errors of 5–10%).

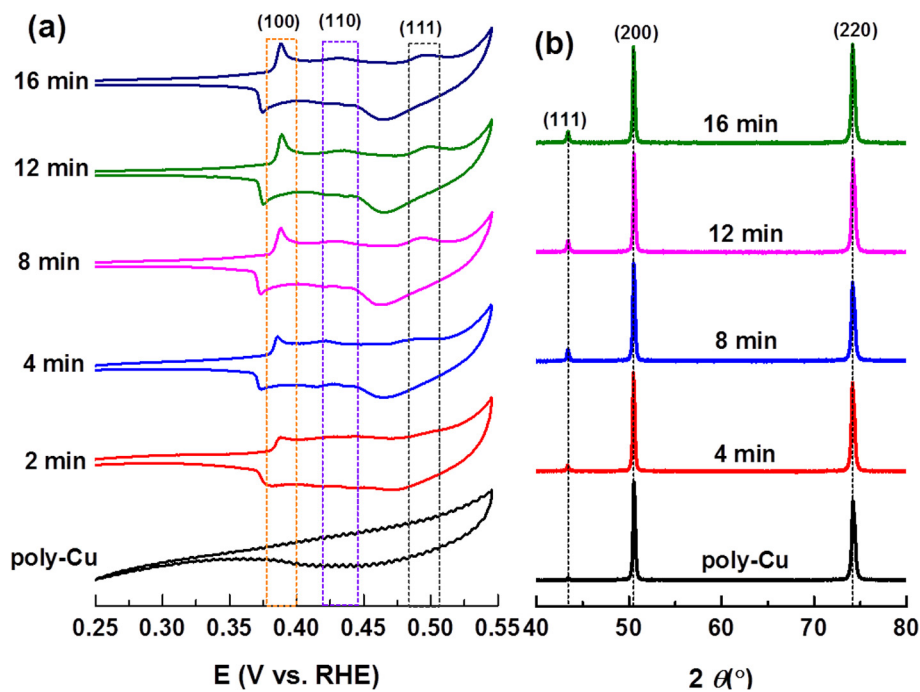
In order to get kinetic insights into the enhanced CO<sub>2</sub> reduction efficiency on the anodized Cu electrode, the partial current density for each major product was plotted against applied potentials (Fig. S7(a) and (d)). Although the H<sub>2</sub> FE on anodized Cu electrode decreased significantly in the potential range of  $-0.3$  to  $-0.75$  V vs. RHE, the partial current densities for H<sub>2</sub> formation were relatively constant, indicating that the increase in the CO<sub>2</sub> reduction selectivity was not due to the decrease in the H<sub>2</sub> evolution rate, but the more rapidly increasing production rates of HCOOH and CO. In comparison, the H<sub>2</sub> formation rate dominated over CO and HCOOH production rates for poly-Cu in the same potential range. In the Tafel plots (Fig. 2(c) and (d)), the anodized Cu electrode shows a slope of 176 and 149 mV dec<sup>-1</sup> for CO and HCOOH production, respectively, which are lower than those of the poly-Cu (236 and 175 mV dec<sup>-1</sup>). Moreover, a higher Tafel slope of 331 mV dec<sup>-1</sup> for H<sub>2</sub> evolution was observed on anodized Cu electrode as compared to poly-Cu (260 mV dec<sup>-1</sup>). These

observations indicate that the anodized Cu surfaces is more favorable for the formation of the CO<sub>2</sub><sup>-</sup> intermediate at less negative applied potentials while suppressing H<sub>2</sub>O reduction, resulting in an enhanced activity and selectivity for the CO<sub>2</sub> reduction [9,10].

To further evaluate the effects of the changes in surface structures on the catalytic properties, the electroadsorption of oxygenated species (e.g., O and/or OH) on anodized Cu surfaces was investigated to identify the exposed facets (Figs. 3 (a) and S8). The CV curve of poly-Cu is featureless due to its extremely low ECSA. In contrast, all the anodized Cu electrodes show three oxidation peaks that can be assigned to the electroadsorption features of oxygenated species on low-index facets of *fcc* Cu [14,17]. Moreover, the peak intensity enhancement for the (100) and (111) facets is even more pronounced in comparison to that of the (110) facet (Fig. 3(a)). Consistent with the literature results [7], these observations clearly suggest that the enhanced activity and selectivity of the anodized Cu



**Fig. 2** (a) FEs for  $\text{H}_2$ , CO and HCOOH vs. applied potential in bulk electrolysis (with experimental errors of 5–10%). (b) Total current density vs. applied potential. (c, d)  $\text{H}_2$ , CO, and HCOOH Tafel plots.



**Fig. 3** (a) CVs of the poly-Cu and the anodized Cu electrodes after electroreduction at  $-0.64$  V vs. RHE. The CVs were recorded in 1 M KOH with a scan rate of 20 mV/s. (b) Normalized XRD patterns of anodized Cu electrodes.

for the CO<sub>2</sub> reduction to CO and HCOOH are associated with the formation of more low-index Cu facets, especially (100) and (111). More (111) facets are favorable for the production of HCOOH at a lower overpotential, in agreement with the results that the HCOOH FE increases with the increasing amount of the (111) facet [7]. Moreover, all the anodized Cu electrodes show an enhanced peak intensity for the (111) facet (Fig. 3(b)), as compared to poly-Cu. Moreover, the intensity ratios of (111) to (200) for anodized Cu electrodes were plotted as a function of the HCOOH FE (Fig. S9). The HCOOH FE shows a positive correlation with the peak intensity ratio of (111) to (200). We thus attribute the improved HCOOH FE of the anodized Cu electrodes to the selectively preserved (111) surface during the anodization-reduction process.

#### 4. Conclusion

A high surface area Cu electrode was prepared by a facile anodization-reduction process. An optimized anodized Cu electrode exhibited a lower onset potential and a 27% CO FE and 37% HCOOH FE at  $-0.64$  V vs. RHE for the CO<sub>2</sub> reduction. The enhanced performance of the anodized Cu towards CO<sub>2</sub> reduction can be attributed to the high electrochemically active surface areas and the favorable formation of low-index Cu facets such as (100) and (111) facets. We have demonstrated that exploring the metal redox electrochemistry is a promising strategy and potentially useful for the preparation of highly efficient metal electrocatalysts for the low overpotential CO<sub>2</sub> reduction.

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#### Appendix A. Supplementary data

Supplementary data associated with this article can be found, in the online version, at <http://dx.doi.org/10.1016/j.jscs.2017.03.003>.

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